

Contents lists available at ScienceDirect

Journal of Solid State Chemistry



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Tunable photoluminescence of NaYF₄:Eu nanocrystals by Sr²⁺ codoping

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ARTICLE INFO

Article history: Received 7 July 2010 Received in revised form 8 October 2010 Accepted 25 October 2010 Available online 30 October 2010

Keywords: Nanocrystals NaYF₄ Rare earth Tunable luminescence

1. Introduction

ABSTRACT

NaYF₄:Eu/Sr nanocrystals were synthesized by a hydrothermal method. Tunable photoluminescence of the NaYF₄:Eu nanocrystals was successfully achieved by codoping with Sr²⁺ ions. With increasing Sr²⁺ concentration, not only the X-ray diffraction peaks of the nanocrystals become broader, but also the positions of them shift toward larger lattice parameters. Eu³⁺ and Eu²⁺ have been found to coexist in an NaYF₄:Eu/Sr. The Eu³⁺/Eu²⁺ emission intensity ratio changed with the Sr²⁺ concentration and excitation wavelength. More interestingly, the spectral configurations of Eu²⁺ and Eu³⁺ also varied with the excitation wavelength, indicating that the nanocrystals have multiple luminescence centers or emitting states.

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In recent years, a great deal of research effort has been devoted to the synthesis of nanocrystals doped with rare earth (RE) ions driven by the fact that doped nanocrystalline phosphors have potential applications in lasers, optical telecommunication, optoelectronic devices, diagnostics, and biological labels [1–14]. Nanosized phosphors or optoelectronic devices usually exhibit novel capabilities, such as tunable wavelengths, high efficiencies, and rapid responsibilities [15–17], resulting from the quantum effects and a high surface-to-volume ratio compared to their bulk counterparts.

It is well known that fluorides, such as NaYF₄, LaF₃, YF₃, and CaF₂, are efficient hosts for luminescent centers, due to their low phonon energies and optical transparency over a wide wavelength range [18–25]. Recently, Chen et al. have reported that NaGdF₄ is also an attractive host for phosphors with interesting upconversion luminescent properties [26]. Even so, NaYF₄ is still acknowledged as one of the most efficient host lattices for *RE* ions.

The luminescence properties of *RE* ions could be improved by changing the lattice parameters of host [27]. The ionic radius of Sr^{2+} is larger than that of Ln^{3+} , and thus, some superior features can be expected in NaYF₄: Ln^{3+} nanocrystals by codoping with Sr^{2+} ions. In this study, NaYF₄:Eu/Sr nanocrystals have been synthesized, using a solution–liquid–solid (LSS) technique. Eu³⁺ and Eu²⁺ have been found to coexist in the Eu/Sr-codoped NaYF₄ nanocrystals. The Eu³⁺/Eu²⁺ emission intensity ratio changed with the Sr^{2+} concentration as well as the excitation wavelength. Spectral analysis indicates that the material has multiple luminescence centers or emitting states.

2. Experiments

In a typical synthesis, 1.5 mL of an NaF aqueous solution (0.5 mol/L) and 0.5 mL of nitrate $[Y(NO_3)_3, Eu(NO_3)_3, and Sr(NO_3)_2]$ aqueous solution (0.5 mol/L) were added to a mixture of NaOH (1.2 g), ethanol (10 mL), deionized water (8 mL), and oleic acid (20 mL), and the solution was thoroughly stirred. Subsequently, the milky colloidal solution was transferred to a 50 mL Teflon-lined autoclave, and heated at 160 °C for 14 h. The systems were then allowed to cool to room temperature. The final products were collected by means of centrifugation, washed with ethanol, and finally dried in vacuum at 80 °C for 4 h.

Characterization: the crystal structure was analyzed by a Rigaku RU-200b X-ray powder diffractometer (XRD), using a nickelfiltered Cu K α radiation (λ =1.5418 Å). The size and morphology of the final products were characterized with a Hitachi H-800 transmission electron microscope (TEM), operated at 200 kV. Samples were prepared by placing a drop of dilute cyclohexane dispersion of the nanocrystals on the surface of a copper grid. The luminescence spectra were recorded with a Hitachi F-4500 fluorescence spectrophotometer at room temperature. For comparison of the luminescence properties of different samples, the luminescence spectra were measured with the same instrument parameters (5 nm for slit width and 400 V for PMT voltage).

3. Results and discussion

3.1. Crystal structure and morphology

Fig. 1 shows the XRD patterns of the as-prepared NaYF₄:Sr/Eu nanocrystals with different Sr^{2+} concentration. It is noted that the

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^{0022-4596/\$ -} see front matter \circledcirc 2010 Elsevier Inc. All rights reserved. doi:10.1016/j.jssc.2010.10.030

concentration refers to the initial Sr^{2+} concentration of reactant. The diffraction peaks in curve 1*a* can be indexed to pure cubic phase NaYF₄ (JCPDS 06-0342). The very weak diffraction peaks marked by asterisks arise from the hexagonal phase NaYF₄ (JCPDS 16-0334). The positions of all diffraction peaks of the nanocrystals can be observed to shift toward larger lattice parameters with increasing Sr^{2+} concentration, indicating the lattice expansion as a result of Sr^{2+} incorporation. Although the valences of Y^{3+} and Sr^{2+} are different, they can be



Fig. 1. XRD patterns of NaYF₄:Sr/Eu nanocrystals with different Sr²⁺ concentration (10% Eu): (a) 0%, (b) 1%, (c) 5%, (d) 15%, (e) 30%, (f) 40%, (g) 50%, (h) 70%, and (i) 80%.

replaced each other with the help of charge compensation. This result is similar to the SrF_2-YF_3 or CaF_2-YF_3 systems reported by Achary et al. [28]. In addition, from bottom to top in Fig. 1, the XRD patterns become broader and broader, suggesting that the crystalline size gradually decreases with increasing Sr^{2+} concentration. Both NaYF₄ and SrF_2 are cubic phases, and thus, similar crystal structures can be observed in samples with different concentration.

The morphologies of the final products were characterized with a Hitachi H-800 transmission electron microscope (TEM) operated at 200 kV. Fig. 2a shows the TEM image of NaYF₄:Sr(30%)/Eu(10%) nanocrystals. They are monodisperse, with a diameter of ~ 10 nm. The high-resolution transmission electron microscope (HRTEM) image of a single nanoparticle is shown in Fig. 2b. The measured lattice spacing is about 0.323 nm, which reveals the single-crystal nature of the product. Fig. 2c shows the TEM image of NaYF₄:Sr(70%)/Eu(10%) nanocrystals. The mean diameter of nanoparticles is \sim 5nm. Obviously, the particle size of an NaYF₄:Sr(70%)/ Eu(10%) is less than that of NaYF₄:Sr(30\%)/Eu(10\%) nanocrystals, which is in accordance with the XRD analysis. During the HRTEM measurement, the elemental components of the nanocrystals were detected by an energy-dispersive X-ray analysis (EDXA). The EDXA result confirmed that the elemental components are Sr, Y, Eu, Na, and F (Fig. 2d). The actual Sr²⁺ concentration is almost consistent with the initial dopant concentration of the reactant.

3.2. Luminescence characteristics

3.2.1. Tunable luminescence of Eu^{2+}/Eu^{3+} in NaYF₄ nanocrystals by Sr²⁺ codoping

Fig. 3 shows the emission spectra of NaYF₄:Sr/Eu nanocrystals under 396 nm excitation, which corresponds to the ${}^{7}F_{0} \rightarrow {}^{5}L_{6}$ transition of Eu³⁺. The emissions of Eu³⁺ and Eu²⁺ were observed simultaneously. The results demonstrate the coexistence of Eu²⁺ and Eu³⁺ ions in the samples. The emission peaks located in the



Fig. 2. (a) TEM and (b) HRTEM images of NaYF₄:Sr(30%)/Eu(10%) nanocrystals. (c) TEM image of NaYF₄:Sr(70%)/Eu(10%) nanocrystals. (d) An EDXA pattern of NaYF₄:Sr(30%)/Eu(10%) nanocrystals.



Fig. 3. Emission spectra (λ_{ex} = 396 nm) of NaYF₄:Sr/Eu nanocrystals with different Sr²⁺ concentration (10% Eu): (a) 0%, (b) 1%, (c) 5%, (d) 15%, (e) 30%, (f) 40%, (g) 50%, (h) 70%, and (i) 80%.

575–630 nm range can be attributed to the ${}^{5}D_{0} \rightarrow {}^{7}F_{1}$ and ${}^{5}D_{0} \rightarrow {}^{7}F_{2}$, which are typical magnetic and electric dipole–dipole transitions of Eu³⁺, respectively. In a site with an inversion symmetry the ${}^{5}D_{0} \rightarrow {}^{7}F_{1}$ is dominating, while in a site without an inversion symmetry, the ${}^{5}D_{0} \rightarrow {}^{7}F_{2}$ is the strongest. Therefore, the intensity ratio of ${}^{5}D_{0} \rightarrow {}^{7}F_{2} - {}^{5}D_{0} \rightarrow {}^{7}F_{1}$ is strongly dependent on the local symmetry of the Eu³⁺ ions. By comparing the emission intensity of the ${}^{5}D_{0} \rightarrow {}^{7}F_{1}$ with that of the ${}^{5}D_{0} \rightarrow {}^{7}F_{2}$ in Fig. 3, it is inferred that the Eu³⁺ ions in NaYF₄:Sr/Eu nanocrystals occupy a site with a small deviation from an inversion symmetry. The peak positions of Eu³⁺ were independent of Sr²⁺ concentration.

In addition to the aforementioned emissions of Eu³⁺, a very strong emission band centered at ~450 nm was also observed, which corresponds to the transition from the 4f⁶5d to 4f⁷ configuration of Eu²⁺ [29,30]. The Eu²⁺ emission in different hosts lies in characteristically different ranges due to different structures. The Eu³⁺/Eu²⁺ emission intensity ratio increases with increasing Sr²⁺ concentration, up to about 70 mol%, and then decreases.

Fig. 4 shows the emission spectra of NaYF₄:Sr/Eu nanocrystals under 325 nm excitation, which corresponds to the 5d–4f emission of Eu²⁺. The inset of Fig. 4 is the magnification of the spectra in the range 334–600 nm. It is observed that the emission of Eu²⁺ is much stronger than that of Eu³⁺. With the increase of Sr²⁺ concentration, this broadband emission of Eu²⁺ experiences a slight bathochromic shift first, then the inverse tendency.

Fig. 5 shows the excitation (monitored at 592 nm) spectra of NaYF₄:Sr/Eu nanocrystals, which consist of the characteristic excitation lines of Eu³⁺. These excitation lines can be clearly assigned: $^{7}F_{0} \rightarrow ^{5}H_{6}$ (~320 nm), $^{7}F_{0} \rightarrow ^{5}D_{4}$ (~364 nm), $^{7}F_{0} \rightarrow ^{5}G_{2}$ (~382 nm), $^{7}F_{0} \rightarrow ^{5}L_{6}$ (~396 nm), $^{7}F_{0} \rightarrow ^{5}D_{2}$ (~467 nm), $^{7}F_{0} \rightarrow ^{5}D_{1}$ (~537 nm), and $^{7}F_{0} \rightarrow ^{5}D_{0}$ (~565 nm). Fig. 6 shows the excitation (monitored at 452 nm) spectra of NaYF₄:Sr/Eu nanocrystals with different Sr²⁺ concentration, which consist of the characteristic excitation bands of Eu²⁺. As mentioned above, the Eu²⁺ emission in different structures. These spectral configurations of Eu²⁺ are different, indicating that the local symmetry of the Eu²⁺ ions is changed by codoping with Sr²⁺ ions. It is imperative to point out that the excitation peaks corresponding to the $^{7}F_{0} \rightarrow ^{5}F_{2}$ and $^{7}F_{0} \rightarrow ^{5}H_{6}$ of an Eu³⁺ overlapped with the transitions of Eu²⁺.



Fig. 4. Emission spectra (λ_{ex} =325 nm) of NaYF₄:Sr/Eu nanocrystals with different Sr²⁺ concentration (10% Eu): (a) 0%, (b) 1%, (c) 5%, (d) 15%, (e) 30%, (f) 40%, (g) 50%, (h) 70%, and (i) 80%. An inset shows the normalized emission spectra in the range 334–600 nm.



Fig. 5. Excitation spectra (λ_{em} =592 nm) of NaYF₄:Sr/Eu nanocrystals with different Sr²⁺ concentration (10% Eu): (a) 0%, (b) 1%, (c) 5%, (d) 15%, (e) 30%, (f) 40%, (g) 50%, (h) 70%, and (i) 80%.

addition, the excitation spectra monitored at 426 and 452 nm have similar behaviors.

3.2.2. Tunable luminescence of NaYF₄:Sr(70%)/Eu(10%) nanocrystals by exciting at different wavelengths

Fig. 7a shows the emission spectra of NaYF₄:Sr(70%)/Eu(10%) nanocrystals, when the Eu³⁺ ions were excited. Fig. 7b shows the emission spectra of NaYF₄:Sr(70%)/Eu(10%) nanocrystals, when the Eu²⁺ ions were excited. It is clearly observed that all the spectra consist of the emissions of Eu³⁺ and Eu²⁺. We suggested that this



Fig. 6. Excitation spectra (λ_{em} =452 nm) of NaYF₄:Sr/Eu nanocrystals with different Sr²⁺ concentration (10% Eu): (a) 0%, (b) 1%, (c) 5%, (d) 15%, (e) 30%, (f) 40%, (g) 50%, (h) 70%, and (i) 80%.



Fig. 7. Emission spectra of $NaYF_4{:}Sr(70\%)/Eu(10\%)$ nanocrystals under different excitation wavelength.



Fig. 8. Excitation spectra of NaYF4:Sr(70%)/Eu(10%) nanocrystals monitored at different emission wavelength.

phenomenon can be attributed to the spectral overlapping and energy transfer between Eu^{3+} and Eu^{2+} ions. The Eu^{3+}/Eu^{2+} emission intensity ratio changes with an excitation wavelength. When the Eu^{2+} ions were excited (λ_{ex} =309, 324, 350, and 356 nm), the emissions from Eu^{3+} are very weak. When the Eu^{3+} ions were excited (λ_{ex} =364, 396, 467, 528, 537, and 565 nm), the emissions from Eu^{3+} become stronger. The emission intensities of Eu^{3+} ions were the strongest when the excitation was performed at 396 nm. In addition, the spectral configurations of Eu^{2+} are also changing with excitation wavelength. This is a normal phenomenon when the material has multiple luminescence centers or emitting states.

Fig. 8 shows the excitation spectra of NaYF₄:Sr(70%)/Eu(10%) nanocrystals monitored at different excitation wavelengths. For Eu³⁺ ions (Fig. 8a), the strongest excitation peak of Eu³⁺ is found to be located at 396 nm (⁷F₀→⁵L₆), which is in accordance with the excitation spectrum of cubic NaYF₄:Eu³⁺ nanocrystals reported previously. For Eu²⁺ ions (Fig. 8b), it was observed that two shoulders appeared in the short wavelength side of the excitation band of Eu²⁺, which correspond to the ⁷F₀→⁵F₂ and ⁷F₀→⁵H₆ transitions. The emission band of an Eu²⁺ shifted towards longer wavelength, when a longer wavelength emission was monitored. The positions of the shoulders were independent of excitation wavelength. These spectral phenomena have been observed for all the samples.

4. Conclusion

Eu/Sr-codoped NaYF₄ nanocrystals have been synthesized by an LSS process. The positions of XRD peaks of the nanocrystals shifted towards larger lattice parameters with increasing Sr²⁺ concentration. Furthermore, the particle size decreased with increasing Sr²⁺ concentration. The influence of Sr²⁺ concentration on the luminescence of NaYF₄:Eu/Sr nanocrystals was studied systemically. The emissions of Eu³⁺ and Eu²⁺ were observed simultaneously, revealing that Eu³⁺ and Eu²⁺ coexist in Eu/Sr-codoped NaYF₄ nanocrystals. The Eu³⁺/Eu²⁺ emission intensity ratio not only changed with the Sr²⁺ concentration, but also changed with the excitation wavelength. The spectral configurations of Eu²⁺ and Eu³⁺ also varied with the excitation wavelength, indicating that the material has multiple luminescence centers or emitting states.

Acknowledgments

This work was supported by the State Key Project of Fundamental Research for Nanoscience and Nanotechnology (2011CB932401), the Foundation for Innovative Research Groups of the National Natural Science Foundation of China (20921001), and the National Natural Science Foundation of China (10979032).

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